

Ternary aluminides LnPd_2Al with YPd_2Si -type structure

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Abstract

Attempts to synthesize YPd_2Si -type representatives LnPd_2Al were successful only with $\text{Ln} \equiv \text{Ce, Pr, Nd}$. This orthorhombic structure ($oP16$; space group $Pnma$) is an ordered variant of the Fe_3C type. Only NdPd_2Al undergoes magnetic ordering above 2 K.

Recently, we reported on rare-earth palladium gallides LnPd_2Ga which crystallize in the YPd_2Si type of structure [1]. The orthorhombic YPd_2Si structure is an ordered derivative of the Fe_3C type, which occurs frequently among Ln_3T alloys (where Ln is a rare-earth element (La–Lu, Y) and T is a transition element (Fe, Ru, Os, ..., Ni, ...)). The YPd_2Si structure, which differs somewhat from its parent structure in the axial ratios, is predominant in the silicides LnPd_2Si [2] and germanides LnPd_2Ge [3] and also occurs in the corresponding Pt compounds [4]. Therefore we expected to meet this structure among the aluminides LnPd_2Al and LnPt_2Al . However, in a first attempt none of the Pt compounds and only three of the Pd compounds could be synthesized with this structure. Neither GdPt_2Sn -type nor Heusler-type phases were detected.

Polycrystalline samples of these compounds were prepared by reacting the constituent elements (minimum 3N grade) in an argon arc furnace in the same way as for the gallium compounds [1]. Only with $\text{Ln} \equiv \text{Ce, Pr}$ and Nd (with increasing difficulty in this sequence) did we obtain the YPd_2Si -type phase. The aluminides are similar to the gallides: brittle, with a silvery metallic lustre. The reason for our failure in the case of $\text{Ln} \equiv \text{Sm, Gd}$, etc. could be a low peritectic temperature.

The room-temperature lattice parameters were derived from Guinier patterns with silicon as internal standard (assuming $a_{\text{Si}(2\theta)}/\lambda_{\text{Cu K}\alpha 1} = 3.52511$). The resulting data are listed in Table 1. Intensity calculations with LAZY PULVERIX [5] and the site parameters of YPd_2Si [2] gave a fair agreement with the observed intensities. The influence of the B metal on the unit cell is clearly visible. The larger size of the Al atom than the Ga atom leads to larger c axes and larger unit cell volumes, but to slightly smaller a and b axes, compared with LnPd_2Ga . The lattice parameters in-

TABLE 1. Room-temperature (295 K) lattice parameters, unit cell volumes and X-ray densities of the YPd_2Si -type LnPd_2Al compounds; space group $Pnma$ (No. 62); Pearson symbol $oP16$. Estimated standard deviations of the last digit (excluding systematic errors due to calibration) are added in parentheses

Ln	a (Å)	b (Å)	c (Å)	V (Å ³)	d_x (g cm ⁻³)
Ce	7.4670(7)	7.0007(8)	5.9433(6)	310.7(1)	8.12
Pr	7.4473(7)	6.9967(7)	5.8960(6)	307.2(1)	8.23
Nd	7.4250(7)	6.9935(7)	5.8634(7)	304.5(1)	8.38

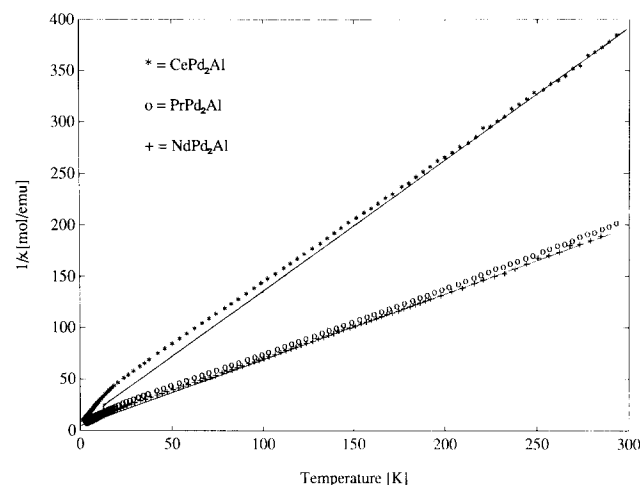


Fig. 1. Temperature dependence of the reciprocal magnetic susceptibilities of CePd_2Al , PrPd_2Al and NdPd_2Al . The straight lines indicate the asymptotic Curie-Weiss law.

dicating a perfect trivalency of Ce in CePd_2Al , as is also the case in CePd_2Ga . Magnetic data show that this trivalency persists down to lower temperatures.

Magnetic measurements were carried out between 1.6 K and room temperature. Below about 200 K in the case of the Ce compound and 150 K in the case of PrPd_2Al and NdPd_2Al the magnetic susceptibility

of all three compounds shows distinct deviations from the Curie–Weiss law due to crystal–electric-field effects (Fig. 1). Asymptotically, the following values are approximated for the effective magneton number n_p and the paramagnetic Curie temperature Θ_p : $2.49 \mu_B$ and -4 K for CePd₂Al, $3.49 \mu_B$ and -8 K for PrPd₂Al and $3.52 \mu_B$ and -4 K for NdPd₂Al. A maximum in the susceptibility curve $\chi(T)$ of NdPd₂Al points to an antiferromagnetic ordering below a Néel temperature $T_N=2.5$ K. No sign of magnetic ordering was detectable in the $\chi(T)$ curves of CePd₂Al and PrPd₂Al.

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